UNIVERSITY OF NOVA GORICA

# MODELLING OF SOLIDIFICATION WITH SOLID PHASE MOVEMENT

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# Outline

- Goals
- Physical phenomena
- Modelling principles
- Physical system
- Derivation of governing equations by volume averaging and mixture theory
- Modeling of microstructure formation
- Coupling of micro and macro equations
- Computational model
- Conclusions
- Literature review



# Goals

- Overview of the literature on modelling of solidification systems
- Physical modelling of solidification systems on the macroscopic scale
- Physical modelling of solidification systems on the microscopic scale
- Development of the multiscale and multiphysics equations that will be used in a Ph. D. project



# Physical phenomena



Melting: solid-liquid



Crystal growth: gas-solid



Solidification: liquid-solid



Allotropic transformations: solid-solid





# Physical phenomena



The grain structure as observed in a directionally solidified Al-7wt%Si casting (left) and solidification process is illustrated for a 2D casting (right). The outer equiaxed, columnar and (inner) equiaxed zones are indicated.



#### Physical phenomena Equiaxed grains growing in a uniform temperature field



(a) Globulitic grains



(b) Dendritic grains



(c) Gray cast iron





(d) Nodular cast iron

Equiaxed grain types frequently encountered in solidification: (a) globulitic grains in Al-1wt%Cu (grain size 100 $\mu$ m), (b) dendritic grains in an Al-7wt%Si alloy (grain size 2mm), (c) gray cast iron, and (d) nodular cast iron (typical graphite nodule size 50 $\mu$ m)

#### Physical phenomena Columnar grains

- Dendrites whose preffered direction is inclined with respect to G must grow faster in order to maintain their relative tip positions
- Tips of the inclined dendrites lie slightly behind those of wellaligned dendrites
- Growth competition among columnar dendrites results in a natural selection of grains with a smal angle  $\phi$



Velocities of (a) eutectic isotherm and eutectic front, and (b) liquidus isotherm and dendrite tips. Because dendrites are constrained to grow along well-defined directions, their velocity, and thus their undercooling, depend on their angle of inclination with respect to the thermal gradient.



# Modeling principles

- Deterministic and stohastic
- Macro level
  - Mixture theory
  - Volume averaging
  - Ensemble averaging
- Micro level
  - Monte Carlo methods
  - Cellular automata method (Point automata)



- Cartesian coordinate system
- Continuum mechanics
- Two-phase, binary alloy system
- Newtonian fluid
- Binary eutectic phase diagram:



- Double diffusive convection: convection depends on temperature and concentration
- Stationary solid phase
- Moving solid phase
- Slurry flow regime
  - $^\circ~$  Volume fraction of the solid phase  $\,g_s\,$  smaller or equal than a predefined packing fraction  $\,g_s^{\,c}\,$
- Porous flow regime
  - $^\circ$  Volume fraction of the solid phase  $g_s$  greater than a predefined packing fraction  $g_s^c$





Example of industrial solidification system: DC casting of aluminium alloys

- Creation of clusters: embryos and nuclei
- Homogeneous nucleation: clusters are assumed to appear spontaneously in a melt free of any impurities
- Heterogeneus nucleation: the clusters form preferentially
- on foreign particles in the melt, or at interfaces (melt-container)
- Nucleation model:

$$\dot{N} = \begin{cases} N_0 \delta(t) & \text{if } T < T_{nucl} \text{ and } \int_0^t \dot{N} dt = 0 \\ N_0 \delta(t) & \text{if } T < T_{nucl} \text{ and } N = 0 \\ 0 & \text{else} \end{cases}$$

 $\Delta T$ 



• Microscopic conservation equations:

Mass:  

$$\frac{\partial}{\partial t} \rho_{k} + \nabla \cdot (\rho_{k} \mathbf{v}_{k}) = 0$$
Momentum:  

$$\frac{\partial}{\partial t} (\rho_{k} \mathbf{v}_{k}) + \nabla \cdot (\rho_{k} \mathbf{v}_{k} \mathbf{v}_{k}) = -\nabla p + \nabla \cdot \mathbf{\tau}_{k} + \mathbf{b}_{k}$$
Energy:  

$$\frac{\partial}{\partial t} (\rho_{k} h_{k}) + \nabla \cdot (\rho_{k} h_{k} \mathbf{v}_{k}) = -\nabla \cdot \mathbf{q}_{k}$$
Species:  

$$\frac{\partial}{\partial t} (\rho_{k} C_{k}) + \nabla \cdot (\rho_{k} C_{k} \mathbf{v}_{k}) = -\nabla \cdot \mathbf{j}_{k}$$

- Microscopic interphase balances:
- Mass balance:

$$(\rho_{\ell} - \rho_{s})\mathbf{v}^{*} \cdot \mathbf{n} = \rho_{\ell}\mathbf{v}_{\ell}^{*} \cdot \mathbf{n} - \rho_{s}\mathbf{v}_{s}^{*} \cdot \mathbf{n}$$

• Momentum balance:

$$-\Gamma_{s}^{\sigma^{*}}+\Gamma_{\ell}^{\sigma^{*}}=-\mathcal{S}_{V}\left\langle 2\gamma_{s\ell}\overline{\kappa}\mathbf{n}\right\rangle ^{*}-\mathcal{S}_{V}\left\langle \nabla_{surf}\gamma_{s\ell}\right\rangle ^{*}$$

• Energy balance (Stefan condition):

$$\boldsymbol{\rho}_{s}L_{f}\mathbf{v}^{*}\cdot\mathbf{n} = \left(k_{s}\frac{\partial T_{s}}{\partial n}\right)^{*} - \left(k_{\ell}\frac{\partial T_{\ell}}{\partial n}\right)^{*}$$

• Solute balance:

$$\rho_{s}\left(1-k_{0K}\right)C_{K\ell}^{*}\left(\mathbf{v}^{*}-\mathbf{v}_{s}^{*}\right)\cdot\mathbf{n}=\rho_{s}D_{Ks}\left(\frac{\partial C_{Ks}}{\partial n}\right)^{*}-\rho_{\ell}D_{K\ell}\left(\frac{\partial C_{K\ell}}{\partial n}\right)^{*}$$



• Microscopic interphase balances:





#### Derivation of governing equations by volume averaging and mixture theory Mixture continuum model

- The development of conservation equations from classical mixture theory is based on the following principles:
  - mixture components may be viewed as isolated subsystems, if interactions with other mixture components are properly treated
  - all properties of the mixture are mathematical consequences of the component properties
  - the mean collective mixture behavior is governed by equations similar to those governing the individual properties
- The partial density of phase k is

$$\overline{\rho}_k = g_k \rho_k$$

• Mixture density and mass averaged velocity:

$$\boldsymbol{\rho} = \sum_{k} \overline{\boldsymbol{\rho}}_{k} \qquad \mathbf{v} = \frac{1}{\boldsymbol{\rho}} \sum_{k} \overline{\boldsymbol{\rho}}_{k} \mathbf{v}_{k}$$

#### Derivation of governing equations by volume averaging and mixture theory Mixture continuum model

• Mass balance:

$$\frac{\partial \rho}{\partial t} + \nabla \cdot (\rho \mathbf{v}) = 0$$

• The continuum equation governing the conservation of momentum can be obtained by summing over each phase

$$\frac{\partial}{\partial t} \left( \sum_{k} \overline{\rho}_{k} \mathbf{v}_{k} \right) + \nabla \cdot \left( \sum_{k} \overline{\rho}_{k} \mathbf{v}_{k} \mathbf{v}_{k} \right) = \nabla \cdot \left( \sum_{k} g_{k} \tau_{k} \right) - \frac{\partial}{\partial x} \left( \sum_{k} g_{k} p_{k} \right) + \sum_{k} \left( \overline{\rho}_{k} B_{k} \right) + F$$

• In a Newtonian fluid, the relation between the shear stress and the velocity gradient is linear, the constant of proportionality being the coefficient of viscosity:

$$\boldsymbol{\tau} = \boldsymbol{\mu} \nabla \mathbf{v}$$



#### Derivation of governing equations by volume averaging and mixture theory Mixture continuum model

• The drag interaction term can be modeled by Darcy's law:

$$f = \frac{\mu_{\ell}}{K} \mathbf{v}$$

• Final momentum equation is:

$$\frac{\partial}{\partial t}(\rho \mathbf{v}) + \nabla \cdot (\rho \mathbf{v} \mathbf{v}) = -\nabla p + \mu_{\ell} \nabla^2 \mathbf{v} - \frac{\mu_{\ell}}{K} \mathbf{v} + \tilde{\rho} \mathbf{g}$$

• Energy balance:

$$\frac{\partial}{\partial t}(\rho h) + \nabla \cdot (\rho h \mathbf{v}) = \nabla \cdot (k \nabla T)$$

• Solute balance:

$$\frac{\partial}{\partial t}(\rho C) + \nabla \cdot (\rho C \mathbf{v}) = \nabla \cdot (\rho D \nabla C)$$

- Consider the volume element which is located in the mushy zone
- The volume element is assumed to be sufficiently large to accurately represent the local structure at the mesoscopic length scale, yet small enough that important variations in the temperature, enthalpy and volume fraction of the solid are resolved for the problem of interest
- We refer to this volume as a 'Representative Volume Element' or RVE



- The macroscopic conservation equations for each phase are obtained by averaging the microscopic equations over the volume,  $V_{\rm R}$
- The volume average of any quantity in phase k is defined as

$$\left\langle \psi \right\rangle = \frac{1}{V_R} \int_{V_R} \psi dV$$

• The averaging process is assumed to satisfy the following properties:

 $\langle \psi + \phi \rangle = \langle \psi \rangle + \langle \phi \rangle \qquad \left\langle \frac{\partial \psi}{\partial x_i} \right\rangle = \frac{\partial \langle \psi \rangle}{\partial x_i}$  $\langle \langle \psi \rangle \phi \rangle = \langle \psi \rangle \langle \phi \rangle \qquad \left\langle \frac{\partial \psi}{\partial t} \right\rangle = \frac{\partial \langle \psi \rangle}{\partial t}$ 

• To be able to average equations over each of the interdispersed phases the phase function  $X_k(\mathbf{x},t)$  is introduced, being equal to one in phase k and zero otherwise,

 $X_{k}(\mathbf{x},t) = \begin{cases} 1 \text{ where } \mathbf{x} \text{ is in phase } k \text{ at time } t \\ 0 \text{ elsewhere} \end{cases}$ 

- The quantity in phase k is expressed as  $\Psi_k = X_k \Psi$
- The phase function  $X_k(\mathbf{x},t)$  is a generalised function and the properties of averaging of derivatives do not apply for it
- The volume average of the phase function is equal to the phase volume fraction

$$\langle X_k \rangle = \frac{1}{V_R} \int_{V_R} X_k dV = \frac{V_k}{V_R} = g_k; \quad \sum_k g_k = 1$$



• The volume average of quantity  $\Psi$  in phase k over the volume  $V_R$  is

$$\langle \psi_k \rangle = \langle X_k \psi \rangle = \frac{1}{V_R} \int_{V_R} X_k \psi dV$$

• The intrinsic volume average, i.e. the average value of the quantitiy inside phase k is

$$\langle \boldsymbol{\psi}_{k} \rangle^{k} = \langle X_{k} \boldsymbol{\psi} \rangle^{k} = \frac{1}{V_{k}} \int_{V_{R}} X_{k} \boldsymbol{\psi} dV = g_{k} \langle \boldsymbol{\psi}_{k} \rangle$$

• The average of the product is

$$\langle \psi_k \phi_k \rangle = \langle X_k \psi \phi \rangle = \langle \psi_k \rangle \langle \phi_k \rangle^k + \langle \hat{\psi}_k \hat{\phi}_k \rangle$$

• The average of the time derivative is

$$\left\langle \left(\frac{\partial \boldsymbol{\psi}}{\partial t}\right)_{k}\right\rangle = \left\langle \frac{\partial \boldsymbol{\psi}}{\partial t}X_{k}\right\rangle = \frac{\partial}{\partial t}\left(g_{k}\left\langle \boldsymbol{\psi}_{k}\right\rangle^{k}\right) + \frac{1}{V_{R}}\int_{A_{k}}\boldsymbol{\psi}_{k}\mathbf{w}_{i}\cdot\mathbf{n}_{k}dA$$

- where  $\mathbf{w}_i$  is the velocity of the phase interface.
- The average of the gradient is

$$\langle \nabla \psi_k \rangle = \nabla \langle \psi_k \rangle - \frac{1}{V_R} \int_{A_k} \psi_k \mathbf{n}_k dA = g_k \nabla \langle \psi_k \rangle^k - \frac{1}{V_R} \int_{A_k} \hat{\psi}_k \mathbf{n}_k dA.$$

• The average of the divergence is

$$\left\langle \left( \nabla \cdot \boldsymbol{\psi} \right)_{k} \right\rangle = \left\langle X_{k} \nabla \cdot \boldsymbol{\psi} \right\rangle = g_{k} \nabla \cdot \left\langle \boldsymbol{\psi}_{k} \right\rangle^{k} + \left\langle \boldsymbol{\psi}_{k} \right\rangle^{k} \cdot \nabla g_{k} - \frac{1}{V_{R}} \int_{A_{k}} \hat{\boldsymbol{\psi}}_{k} \mathbf{n}_{k} dA.$$

- Hypotheses:
  - the densities of both phases are assumed to be equal and constant, except for bouyancy Boussinesq approximation,  $\rho_s = \rho_\ell = \rho$
  - $\,\circ\,\,\,$  laminar flow and constant viscosity  $\,\mu_\ell\,$
  - solute diffusion in the liquid and solid at the macroscopic scale is neglected
  - local thermodynamic equilibrium is assumed, with perfect solute diffusion in both phases (lever rule)
  - the mushy zone is an isotropic porous medium whose permeability K is defined by Carman-Kozeny relation
  - constant thermal conductivity
  - constant heat capacity
  - saturated medium  $(g_s + g_l = 1)$
  - thermal equilibrium between phases (  $\langle T_s \rangle^s = \langle T_\ell \rangle^\ell = T$  )
  - fixed solid phase (  $\langle \mathbf{v}_s \rangle^s = 0$ )



• The averaged mass balance for phase k is

$$\frac{\partial}{\partial t}(g_k) + \nabla \cdot (g_k \langle \mathbf{v}_k \rangle^k) = \frac{1}{\rho} (\Gamma_k + \Phi_k)$$

• The averaged momentum balance for phase k is

$$\frac{\partial}{\partial t} \left( g_k \left\langle \mathbf{v}_k \right\rangle^k \right) + \nabla \cdot \left( g_k \left\langle \mathbf{v}_k \right\rangle^k \left\langle \mathbf{v}_k \right\rangle^k \right) = -\frac{g_k}{\rho} \nabla p_k + \frac{1}{\rho} \nabla \cdot \left( g_k \left\langle \tau_k \right\rangle^k \right) \\ + \frac{g_k}{\rho} \rho_k^b \mathbf{g} + \frac{1}{\rho} \left( \mathbf{M}_k^d + \mathbf{M}_k^{\Gamma} + \mathbf{M}_k^{\Phi} \right)$$

• Phase change and grain nucleation are usually neglected

- The averaged solute mass balance for species *i* in phase *k* is  $\frac{\partial}{\partial t} \left( g_k \left\langle C_k^i \right\rangle^k \right) + \nabla \cdot \left( g_k \left\langle C_k^i \right\rangle^k \left\langle \mathbf{v}_k \right\rangle^k \right) = \frac{1}{\rho} \left( J_k^{i,j} + J_k^{i,\Gamma} + J_k^{i,\Phi} \right)$
- The averaged heat balance for phase k is

$$\frac{\partial}{\partial t} \left( g_k \left\langle h_k \right\rangle^k \right) + \nabla \cdot \left( g_k \left\langle h_k \right\rangle^k \left\langle \mathbf{v}_k \right\rangle^k \right) = \frac{1}{\rho} \nabla \cdot \left( g_k k_k \left\langle \nabla T_k \right\rangle^k \right) \\ + \frac{1}{\rho} \left( Q_k^j + Q_k^\Gamma + Q_k^\Phi \right)$$

- The interphase transfer of mass and solute and nucleation require aditional modeling
- This information has been lost during the volume averaging process

- The nucleation and phase change terms  $(\mathbf{M}_k^{\Phi} \approx 0 \text{ and } \mathbf{M}_k^{\Gamma} \approx 0)$  are neglected
- The liquid phase is modeled as a Newtonian fluid and the tangential stress term is thus  $\nabla \cdot (g_{\ell} \langle \boldsymbol{\tau}_{\ell} \rangle^{\ell}) = \nabla \cdot (\mu_{\ell} \nabla g_{\ell} \langle \mathbf{v}_{\ell} \rangle^{\ell})$
- The interfacial momentum balance  $(\mathbf{M}_{\ell}^{d} = -\mathbf{M}_{s}^{d})$  is applied and the phase change term is neglected  $(\mathbf{M}_{\ell}^{\Gamma} \approx 0)$
- The solid phase momentum  $(-g_s \nabla p + \mathbf{M}_s^d + g_s \mathbf{g} \rho_s^b = 0)$  can be applied and we obtain

$$\frac{\partial}{\partial t} \left( \rho g_{\ell} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} \right) + \nabla \cdot \left( \rho g_{\ell} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} \right) = -\nabla p + \nabla \cdot \left( \mu_{\ell} \nabla \left( g_{\ell} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} \right) \right) + \tilde{\rho} \mathbf{g}$$



• In the liquid momentum balance the interfacial drag force is modeled according to the Darcy model:

$$\mathbf{M}_{k}^{d} = -\frac{g_{\ell}^{2} \boldsymbol{\mu}_{\ell}}{K} \langle \mathbf{v}_{\ell} \rangle^{\ell}$$

• The momentum balance in the liquid phase in the porous region is thus

$$\frac{\partial}{\partial t} \left( \rho g_{\ell} \langle \mathbf{v}_{\ell} \rangle^{\ell} \right) + \nabla \cdot \left( \rho g_{\ell} \langle \mathbf{v}_{\ell} \rangle^{\ell} \langle \mathbf{v}_{\ell} \rangle^{\ell} \right) = -g_{\ell} \nabla p + g_{\ell} \mu_{\ell} \nabla^{2} \langle \mathbf{v}_{\ell} \rangle^{\ell} - \frac{g_{\ell}^{2} \mu_{\ell}}{K} \langle \mathbf{v}_{\ell} \rangle^{\ell} + g_{\ell} \tilde{\rho} \mathbf{g}$$

• The Kozeny–Carman equation is used to model the permeability of the columnar structure

$$K = \frac{g_{\ell}^2}{\left(1 - g_{\ell}\right)^2} \frac{d_g^2}{\pi^2 k_K \tau^2}$$

• In order to get the solution, we have to solve a set of equations

$$\nabla \cdot \langle \mathbf{v}_{\ell} \rangle^{\ell} = 0,$$

$$\frac{\partial}{\partial t} \Big( \rho g_{\ell} \langle \mathbf{v}_{\ell} \rangle^{\ell} \Big) + \nabla \cdot \Big( \rho g_{\ell} \langle \mathbf{v}_{\ell} \rangle^{\ell} \Big) = -g_{\ell} \nabla p + g_{\ell} \mu_{\ell} \nabla^{2} \langle \mathbf{v}_{\ell} \rangle^{\ell} - \frac{g_{\ell}^{2} \mu_{\ell}}{K} \langle \mathbf{v}_{\ell} \rangle^{\ell} + g_{\ell} \tilde{\rho} \mathbf{g},$$

$$\rho \frac{\partial \langle h \rangle}{\partial t} + \nabla \cdot \Big( \rho c_{p} \nabla T^{\ell} \cdot \langle \mathbf{v}_{\ell} \rangle^{\ell} \Big) - \nabla \cdot (k \nabla T) = 0,$$

$$\rho \frac{\partial \langle C \rangle}{\partial t} + \nabla \langle C_{\ell} \rangle^{\ell} \cdot \langle \mathbf{v}_{\ell} \rangle^{\ell} = 0$$

- Density variation with temperature and composition:  $\tilde{\rho} = \rho \left( 1 - \beta_T (T - T_0) - \beta_C \left( \langle C_\ell \rangle^\ell - C_0 \right) \right)$
- Enthalpies:

 $\langle h_s \rangle^s = c_p T \qquad \langle h_\ell \rangle^\ell = c_p T + L \qquad \langle h \rangle = c_p T + g_\ell L$ 

• Microsegregation model (lever rule):

 $\left\langle C\right\rangle = g_{\ell} \left\langle C_{\ell}\right\rangle^{\ell} + g_{s} \left\langle C_{s}\right\rangle^{s} = \left(g_{\ell} + k_{p} \left(1 - g_{\ell}\right)\right) \left\langle C_{\ell}\right\rangle^{\ell} \qquad T = T_{f} + m \left\langle C_{\ell}\right\rangle^{\ell}$ 

• The initial and boundary conditions:

 $C = C_0$   $\frac{dT}{d\mathbf{n}} = 0$   $\mathbf{v} = 0$  $\langle \mathbf{v}_{\ell} \rangle^{\ell} (x = W, t) = \langle \mathbf{v}_{\ell} \rangle^{\ell} (y = H, t) = \langle \mathbf{v}_{\ell} \rangle^{\ell} (y = 0, t) = 0$  $\frac{\partial}{\partial x} \left\langle \mathbf{v}_{\ell y} \right\rangle^{\ell} \left( x = 0, t \right) = 0$  $\frac{\partial}{\partial r}T(x=W,t) = \frac{q}{\lambda}(T-T_{ext})$ W  $\frac{dT}{d\mathbf{n}} = 0$   $\frac{dC}{d\mathbf{n}} = 0$ Symmetry line  $\frac{dV_y}{d\mathbf{n}} = 0$  $\frac{\partial}{\partial x}T(x=0,t) = \frac{\partial}{\partial y}T(y=0,t) = \frac{\partial}{\partial y}T(y=H,t) = 0$  $C = C_0$  $\frac{\partial}{\partial r} \langle C \rangle (x = 0, t) = 0$  $\frac{dT}{d\mathbf{n}} = \frac{q}{\lambda} \left( T - T_{ext} \right)$ Η  $\langle C \rangle (x = W, t) = \langle C \rangle (y = H, t) = \langle C \rangle (y = 0, t) = C_0$  $\mathbf{v} = \mathbf{0}$  $v_x = 0$  $\langle \mathbf{v}_{\ell} \rangle^{\ell} (\mathbf{p}, t=0) = 0$  $T(\mathbf{p},t=0)=T_0$ x  $\langle C \rangle (\mathbf{p}, t=0) = C_0$ Mushv Solid Liquid y  $C = C_0$   $\frac{dT}{d\mathbf{n}} = 0$   $\mathbf{v} = 0$ 



# Derivation of governing equations by volume averaging and mixture theory Comparison of momentum equations

• Mixture continuum momentum equation:

$$\frac{\partial}{\partial t}(\rho \mathbf{v}) + \nabla \cdot (\rho \mathbf{v} \mathbf{v}) = -\nabla p + \mu_{\ell} \nabla^2 \mathbf{v} - \frac{\mu_{\ell}}{K} \mathbf{v} + \tilde{\rho} \mathbf{g}$$

• Volume averaged momentum equation:

$$\frac{\partial}{\partial t} \left( \rho g_{\ell} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} \right) + \nabla \cdot \left( \rho g_{\ell} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} \right) = -g_{\ell} \nabla p + g_{\ell} \mu_{\ell} \nabla^{2} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} - \frac{g_{\ell} \mu_{\ell}}{K} \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} + g_{\ell} \tilde{\rho} \mathbf{g}$$



# Modeling of microstructure formation



A 2D schematic of an equiaxed dendrite surrounded by a cruciform (a), square (b) and spherical envelope (c)



# Modeling of microstructure formation

• Lever rule (equilibrium solidification model): complete mixing of solute in liquid and solid is assumed

$$f_{s} = \frac{1}{1 - k_{0}} \frac{T_{\ell} - T}{T_{m} - T}; \ T_{s} \le T \le T_{\ell} \qquad k_{0} = \frac{C_{s}}{C_{\ell}}$$

• Scheil model: complete mixing of solute in liquid and no mixing of solute in solid is assumed

$$f_{s} = 1 - \left(\frac{T_{m} - T}{T_{m} - T_{\ell}}\right)^{\frac{1}{k_{0} - 1}}; T_{s} \le T \le T_{\ell}$$

# Phase change kinetics

- Solution of temperature, concentration and velocity field on macro level by a meshless method
- The phase change can be achived by undercooling a liquid below its melting temperature
- Equilibrium melting temperature:

$$T_f = T_m + m \left( \left\langle C_\ell \right\rangle^\ell - C_0 \right) - \Gamma \kappa$$

• The interface growth velocity:

$$V_{g}^{*}(\mathbf{p},t) = \mu_{k}\left(T_{f} - T(\mathbf{p},t)\right) + V_{\mathbf{v}}(\mathbf{v}); \ \mathbf{p} = \Gamma_{s,\ell}$$

• Grain growth as a function of velocity, assuming no back diffusion in the solid (Wang and Beckermann, 1996a.):

$$V_{\mathbf{v}}(\mathbf{v}) = \frac{4\sigma^* D_{\ell} m_{\ell} (k_0 - 1) C_{\ell}^*}{\Gamma} \operatorname{Pe}_t^2; \quad C_{\ell}^* = \frac{T - T_m}{m_{\ell}}; \quad \Omega = \frac{C_{\ell}^* - \langle C_{\ell} \rangle^{\ell}}{C_{\ell}^* (1 - k_0)}$$

### Phase change kinetics

- For computational efficiency, the inverse of the analytical solution has been curve fitted
- Diffusion-dominated growth:

$$\operatorname{Pe}_{t} = a \left( \frac{\Omega}{1 - \Omega} \right)^{b}; \ a = 0.4567; \ b = 1.195$$

• Convection-dominated growth:

$$\operatorname{Pe}_{t} = a \left( \frac{\Omega}{1 - \Omega} \right)^{b}; \quad \operatorname{Pe}_{\infty} = \left| \left\langle \mathbf{v}_{\ell} \right\rangle^{\ell} - \left\langle \mathbf{v}_{s} \right\rangle^{s} \right| R_{t} / D_{\ell}$$
$$a = 0.4567 + 0.173 \operatorname{Pe}_{\infty}^{0.55}; \quad b = 1.195 - 0.145 \operatorname{Pe}_{\infty}^{0.16}$$

# Phase change kinetics

• Thermodynamic anisotropy:

$$\Gamma = \overline{\Gamma} \left[ 1 - \delta_t \cos \left( S \left( \theta - \theta_{def} \right) \right) \right]$$

• Kinetic anisotropy:

$$V = V_g^*(\mathbf{p}, t) \left[ 1 - \delta_k \cos\left(S\left(\theta - \theta_{def}\right)\right) \right]$$

### Coupling of micro and macro equations





• The transport equations for each phase are averaged separately and then summed up by taking into account the interphase balances

Macro scale equations:

 $\mathcal{R} = \begin{cases} 1 & g_s > g_s^c \\ 0 & g_s \le g_s^c \end{cases}$  $\frac{\partial n}{\partial t} + \nabla \cdot \left( \mathbf{v}_{s} n \right) = \dot{N}$  $\mathbf{v}_r = \frac{2}{9} \frac{\left(\rho_s - \rho_\ell\right)}{\mu} \mathbf{g} d^2$  $\mathbf{v} = f_{\ell} \mathbf{v}_{\ell} + f_{\epsilon} \mathbf{v}_{\epsilon}$  $\rho = g_{\ell} \rho_{\ell} + g_{s} \rho_{s}$  $K = K_0 \frac{f_\ell^3}{\left(1 - f_\ell\right)^2}$  $g_k = \frac{\rho}{\rho_k} f_k$  $\mathbf{b} = \boldsymbol{\rho}_{ref} \left[ 1 - \boldsymbol{\beta}_T (T - T_{ref}) - \boldsymbol{\beta}_C (C_\ell - C_{ref}) \right] \mathbf{g}$  $f_{s} + f_{\ell} = 1; g_{s} + g_{\ell} = 1$  $h = c_n T + f_\ell L$  $\nabla \cdot \mathbf{v} = 0$  $N_0 \delta(t)$  if  $T < T_{nucl}$  and  $\int_0^t \dot{N}^{\Phi} dt = 0$  $\rho \frac{\partial \mathbf{v}}{\partial t} + \frac{\rho}{f_{\ell}} (\nabla \mathbf{v}) \mathbf{v} = -f_{\ell} \nabla p + \mu \nabla^2 \mathbf{v} - \mathcal{R} \left( f_{\ell} \frac{\mu}{K} \mathbf{v} \right)$  $\dot{N} = \left\{ N_0 \delta(t) \text{ if } T < T_{nucl} \text{ and } N = 0 \right\}$  $-(1-\mathcal{R})\nabla \cdot (\rho f_s f_\ell \mathbf{v}_r \mathbf{v}_r) + f_\ell \mathbf{b}$ else  $\rho \frac{\partial h}{\partial t} + \rho \mathbf{v} \cdot \nabla h = \lambda \nabla^2 T$  $\frac{\partial C}{\partial t} + \mathbf{v} \cdot \nabla C_{\ell} = 0$ 



• The transport equations for each phase are averaged separately and then summed up by taking into account the interphase balances

Micro scale equations:

$$T_{f} = T_{m} + m_{\ell} (C_{\ell} - C_{0}) - \Gamma \kappa$$

$$V_{g}^{*}(\mathbf{p}, t) = \mu_{k} (T_{f} - T(\mathbf{p}, t)) + V_{\mathbf{v}}(\mathbf{v}); \mathbf{p} = \Gamma_{s,\ell}$$

$$V_{\mathbf{v}}(\mathbf{v}) = \frac{4\sigma^{*} D_{\ell} m_{\ell} (k_{0} - 1) C_{\ell}^{*}}{\Gamma} \operatorname{Pe}_{t}^{2}$$

$$C_{\ell}^{*} = \frac{T - T_{m}}{m_{\ell}}$$

$$\Omega = \frac{C_{\ell}^{*} - \langle C_{\ell} \rangle^{\ell}}{C_{\ell}^{*} (1 - k_{0})}$$

$$\Gamma = \overline{\Gamma} \Big[ 1 - \delta_{t} \cos \Big( S \big( \theta - \theta_{def} \big) \Big) \Big]$$

$$V = V_{g}^{*}(\mathbf{p}, t) \Big[ 1 - \delta_{k} \cos \Big( S \big( \theta - \theta_{def} \big) \Big) \Big]$$

• The initial and boundary conditions:

 $\mathbf{v}(x=W,t) = \mathbf{v}(y=H,t) = \mathbf{v}(y=0,t) = 0$   $\frac{\partial}{\partial x}\mathbf{v}_{y}(x=0,t) = 0$   $\frac{\partial}{\partial x}T(x=W,t) = \frac{q}{\lambda}(T-T_{ext})$   $\frac{\partial}{\partial x}T(x=0,t) = \frac{\partial}{\partial y}T(y=0,t) = \frac{\partial}{\partial y}T(y=H,t) = 0$   $\frac{\partial}{\partial x}C(x=0,t) = 0$   $C(x=W,t) = C(y=H,t) = C(y=0,t) = C_{0}$   $\mathbf{v}(\mathbf{p},t=0) = 0$   $T(\mathbf{p},t=0) = T_{0}$  $C(\mathbf{p},t=0) = C_{0}$ 



• Comparison between the models (governing equations):

$$\mathcal{R} = \begin{cases} 1 & g_s > g_s^c \\ 0 & g_s \le g_s^c \end{cases}$$

$$\mathbf{v} = g_{\ell} \mathbf{v}_{\ell} + g_{s} \mathbf{v}_{s} \qquad \qquad \mathbf{v} = g_{\ell} \mathbf{v}_{\ell}$$

• Mass balance:  $\nabla \cdot \mathbf{v} = 0$ 

 $\nabla \cdot \mathbf{v} = 0$ 

• Momentum balance:

$$\frac{\partial}{\partial t}(\rho g_{\ell} \mathbf{v}) + \nabla \cdot (\rho g_{\ell} \mathbf{v} \mathbf{v}) = -g_{\ell} \nabla p + g_{\ell} \mu_{\ell} \nabla^{2} \mathbf{v} \qquad \qquad \frac{\partial}{\partial t}(\rho g_{\ell}) = -\mathcal{R}\left(\frac{g_{\ell}^{2} \mu_{\ell}}{K} \mathbf{v}\right) - (1 - \mathcal{R}) \nabla \cdot (\rho g_{s} g_{\ell} \mathbf{v}_{r} \mathbf{v}_{r}) + g_{\ell} \tilde{\rho} \mathbf{g}, \qquad \qquad -g_{\ell} \nabla p - g_{\ell} \nabla$$

$$\frac{\partial}{\partial t} (\rho g_{\ell} \mathbf{v}) + \nabla \cdot (\rho g_{\ell} \mathbf{v} \mathbf{v}) = -g_{\ell} \nabla p + g_{\ell} \mu_{\ell} \nabla^{2} \mathbf{v} - \frac{g_{\ell}^{2} \mu_{\ell}}{K} \mathbf{v} + g_{\ell} \tilde{\rho} \mathbf{g}$$

• Energy balance:

$$\rho \frac{\partial h}{\partial t} + \rho \mathbf{v} \cdot \nabla h = \lambda \nabla^2 T \qquad \qquad \rho \frac{\partial h}{\partial t} + \rho \mathbf{v} \cdot \nabla h = \lambda \nabla^2 T$$

• Species balance:

$$\frac{\partial C}{\partial t} + \mathbf{v} \cdot \nabla C_{\ell} = 0 \qquad \qquad \frac{\partial C}{\partial t} + \mathbf{v} \cdot \nabla C_{\ell} = 0$$

• Comparison between the models:



# Conclusions

- Introduction to physical phenomena in solidification
- Overview of physical modelling of solidification systems on the macroscopic and microscopic scale
- Description of point automata method
- Developed coupling of microscopic and macroscopic equations

#### Future steps:

- Solving the solidification system with movement of the solid phase by LRBFCM
  - Temperatures,
  - Concentrations,
  - Velocities,
  - Pressure,
  - Grain structure, size and concentration
- Comparison with measurements (Fautrelle, Grenoble)

#### Literature overview

- Dantzig, J.A, Rappaz, M. (2009). Solidification. EPFL Press
- Campbell, J. (2003). Casting: The New Metallurgy of Cast Metals. Butterworth-Heinemann
- Cantor, B. and O'Reilly, K (2003). editors. Solidification and Casting: an Oxford-Kobe materials text. CRC Press
- Kurz, W. and Fisher., D. J. (1998). Fundamentals of solidification. Trans. Tech. Publ., Aedermansdorf, Switzerland, 4th edition
- Bennon, W. D. and Incropera, F. P. (1987). A continuum model for momentum, heat and species transport in binary solid-liquid phase change systems – I. Model formulation. *International Journal* of Heat and Mass Transfer, 30:2161-2169
- Flemings, M. C. (1974). Solidification Processing. McGraw-Hill, New York
- Heine, R. W., Loper C. R.,and P. C. Rosenthal (1967). Principles of Metal Casting. McGraw-Hill, New York
- Chalmers, B. (1964). Principles of Solidification. Wiley, New York

#### Literature overview

- Janssens, K., Raabe, D. And Kozeschnik, E (2007). Computational Materials Engineering: An Introduction to Microstructure Evolution. Elsevier Inc., USA
- Mosbah, S. (2008). Multiple Scales Modeling of Solidification Grain Structures and Segregation in Metallic Alloys. PhD Thesis, L'ecole Nationale Superieure des Mines de Paris
- Gandin, C.A, Desbiolles, J.L., Rappaz, M. and Thevoz, P. (1999). A three-dimensional Cellular Automaton Finite Element model for the prediction of solidification grain structures. *Metallurgical and Material Transactions A*, 30A:3153
- Wang, C. Y. and Beckermann, C. (1996a). Equiaxed dendritic solidification with convection: Part I. Multiscale/multiphase modeling. *Metallurgical and Material Transactions A*, 27A(9):2754-2764
- Gandin, C.A. and Rappaz, M. (1994). A coupled Finite Element Cellular Automata model for the prediction of dendritic grain structures in solidification processes. *Acta Met. Mater.*, 42:2233
- Ni, J. and Beckermann, C. (1991). A volume-averaged two-phase model for transport phenomena during solidification. *Metallurgical Transactions B*, 22B:349-361
- Rappaz, M. (1989). Modelling of microstructure formation in solidification processes. Int. Mater. Rev., 34:93

# **THANK YOU FOR YOUR ATTENTION!**

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